



Applications of 1st-Principles Based Evolution Algorithms in Material Design and Discovery

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Outlines:

- ***What is 1st-Principles Simulations?***

Why do we need Evolution Algorithm?

- ***Proton Order/Disorder transitions in Ice***
a 70 year old problem in ice physics.

- ***ZnO based alloy***

Alloy Synthesis on the Cloud ?

- ***Structure of nano-sized clusters***

***Knowing the structure is
the FIRST step toward
understanding Material Properties***



Collaborators and Group Members



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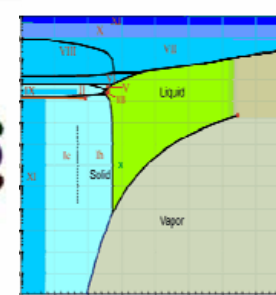
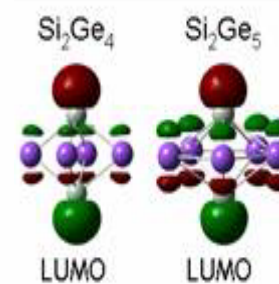
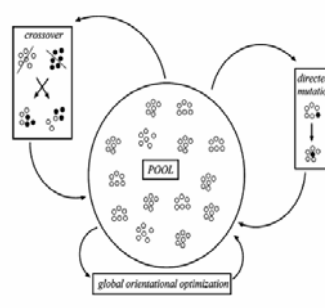
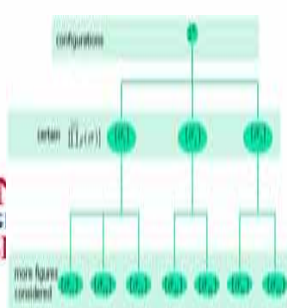
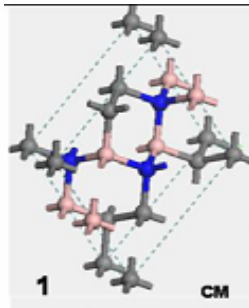
Bing Dan



Zhang Jingyun



Lee Ching-Tao



Basis of 1st-Principles Methods

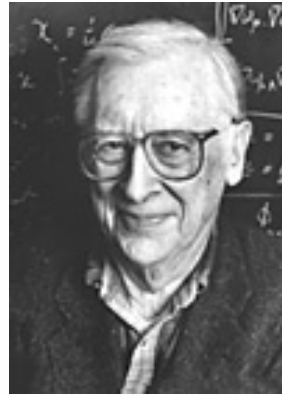
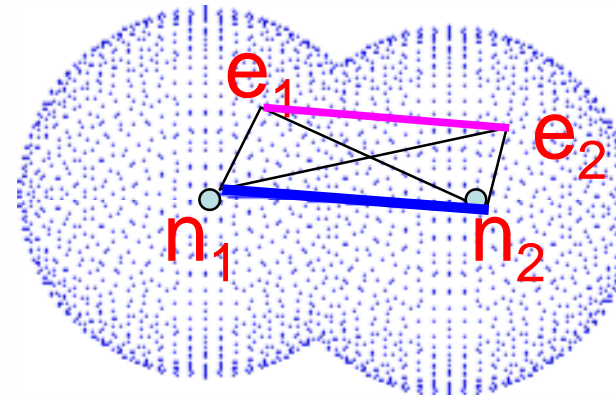


Erwin Schrödinger Paul A. M. Dirac
Nobel Prize in Physics 1933

Dirac (1929): "The underlying physical laws necessary for the mathematical theory of a large part of physics and the whole of chemistry are thus completely known."

Schrödinger equation:

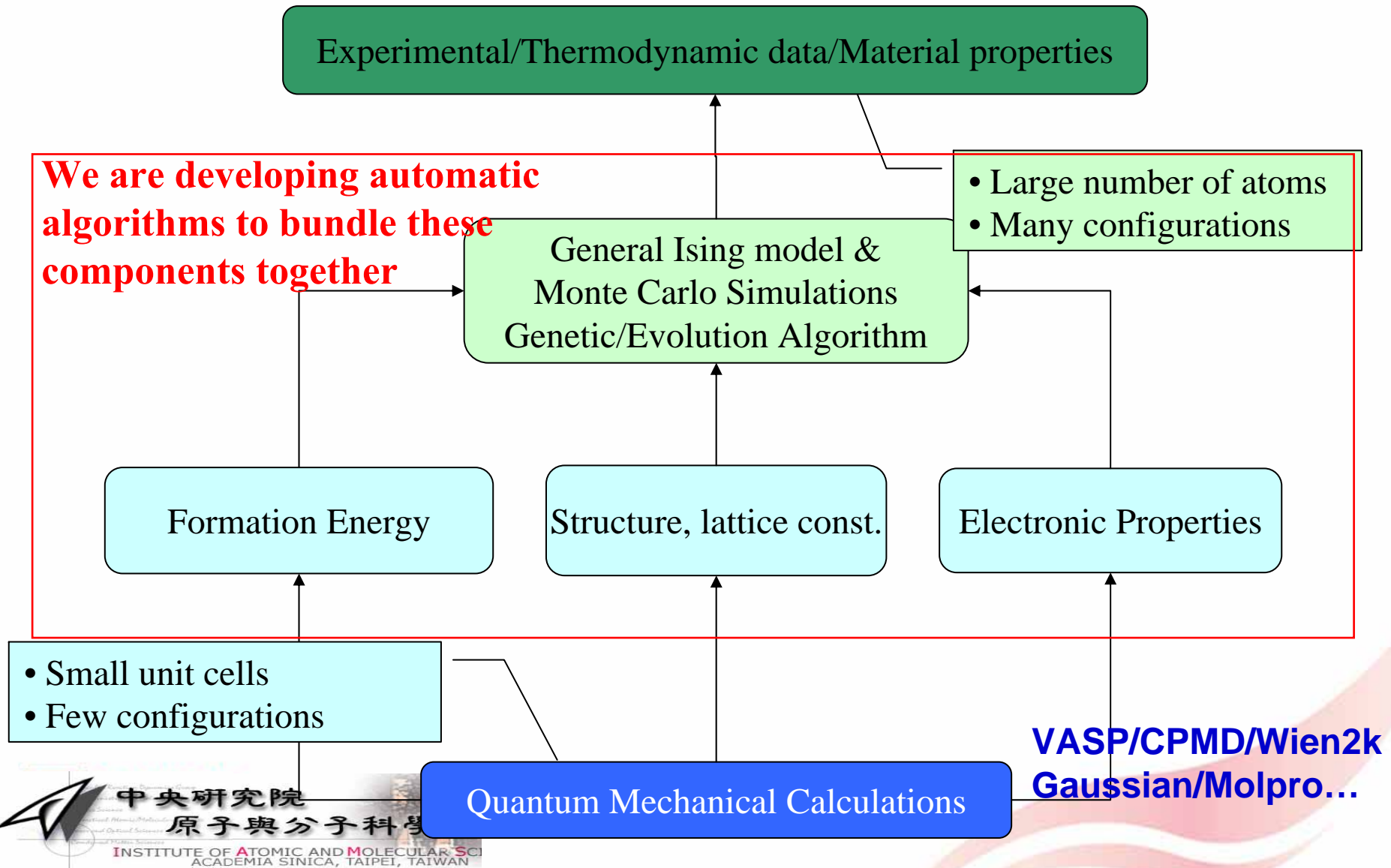
$$H\psi = E\psi$$



Walter Kohn John A. Pople
Nobel Prize in Chem. 1998

"for his development of DFT and comp. methods in quantum chemistry"

Multi-scale simulation from Ab Initio



Material Synthesis on the CLOUD

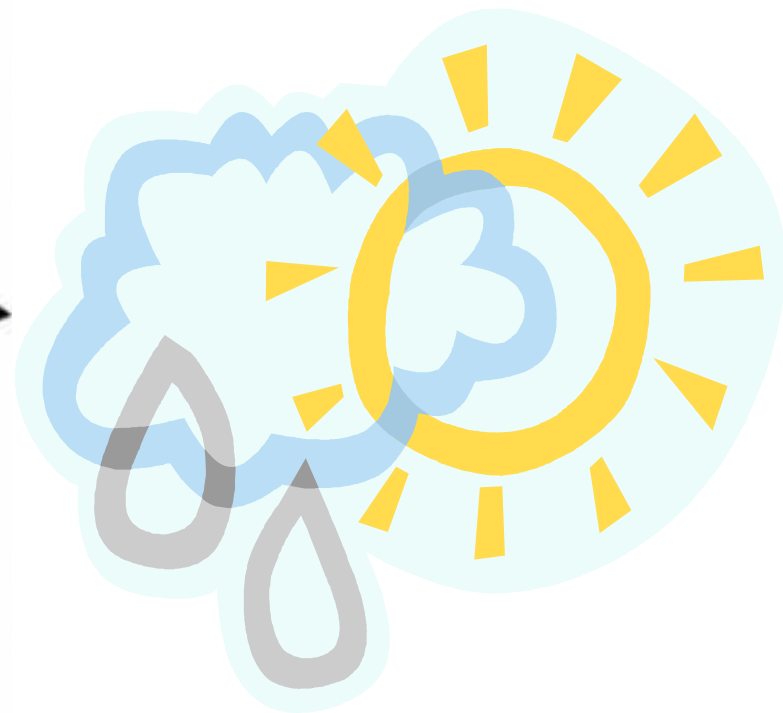
Periodic table of elements with color-coded categories and state indicators. The table is organized into groups and periods. The legend indicates the following categories:

- Alkali metals (Yellow)
- Alkaline earth metals (Orange)
- Transition metals (Red)
- Lanthanide series (Purple)
- Actinide series (Dark Purple)
- Poor metals (Light Blue)
- Nonmetals (Green)
- Noble gases (Cyan)
- Solid (C)
- Liquid (B)
- Gas (H)
- Synthetic (T)

Atomic masses in parentheses are those of the most stable or common isotope.

Note: The following numbers 117 and 118 were added in 2016 by the International Union of Pure and Applied Chemistry. The names of elements 113, 115 and the Lanthanide and Actinide series.

GRID → CLOUD



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Why Multi-scale and go CLOUD?



- *Proton Order/Disorder transitions in Ice*
a 70 year old problem in ice physics.



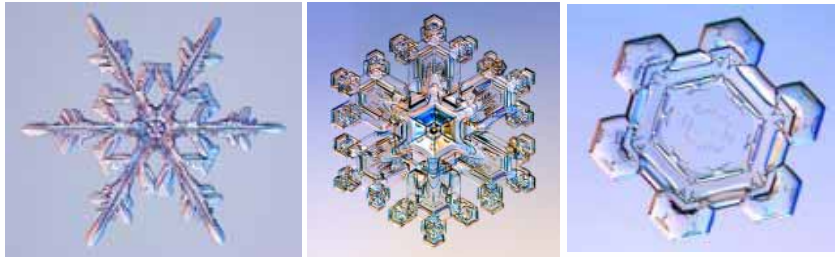
- *Be_xZn_{1-x}O?*
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WHAT!! Ordinary Ice is NOT crystal.

- Snow flakes have beautiful hex-symmetry !!

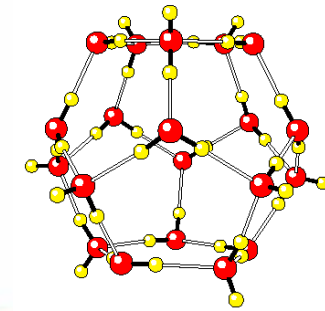
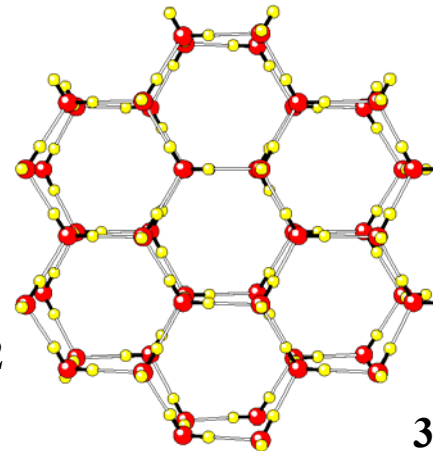


<http://www.its.caltech.edu/~atomic/snowcrystals/>

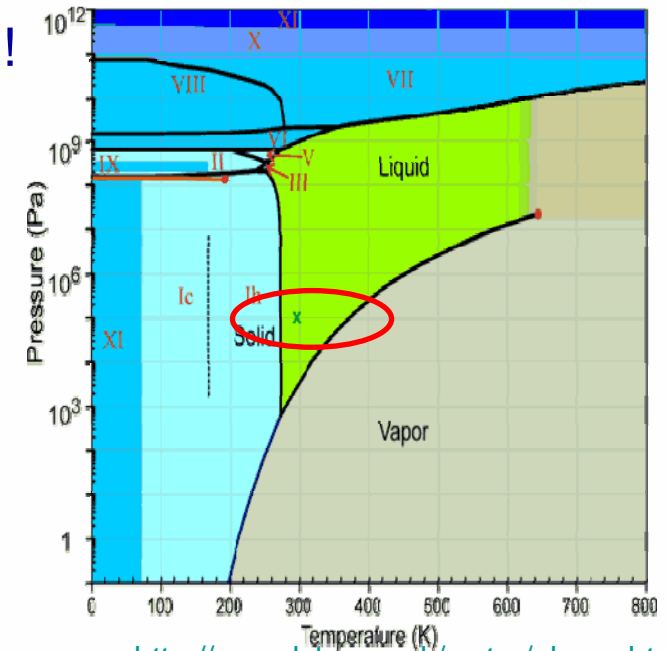
- A **crystal** is a solid in which the components are packed in a regularly ordered, repeating pattern extending in all three spatial dimensions

- Ice has a disordered proton distribution.

In this 48-water unit cell of ice-Ih, there are 2,404,144,962 H-bond isomers.

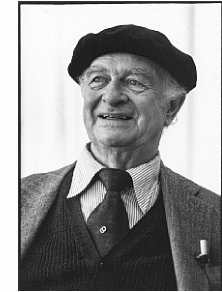


3,600,000 H-bond isomers.



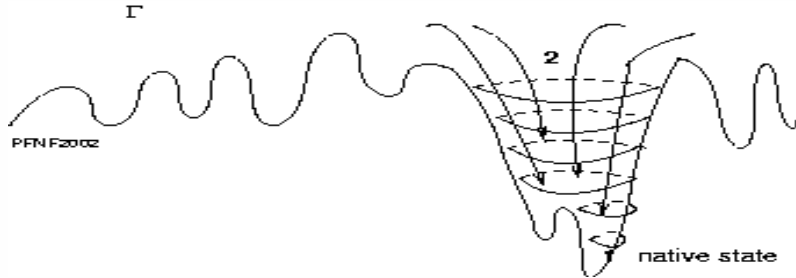
<http://www.lsbu.ac.uk/water/phase.html>

Residual Entropy of ice-Ih



L. Pauling *JACS* 57, 2680 (1935)

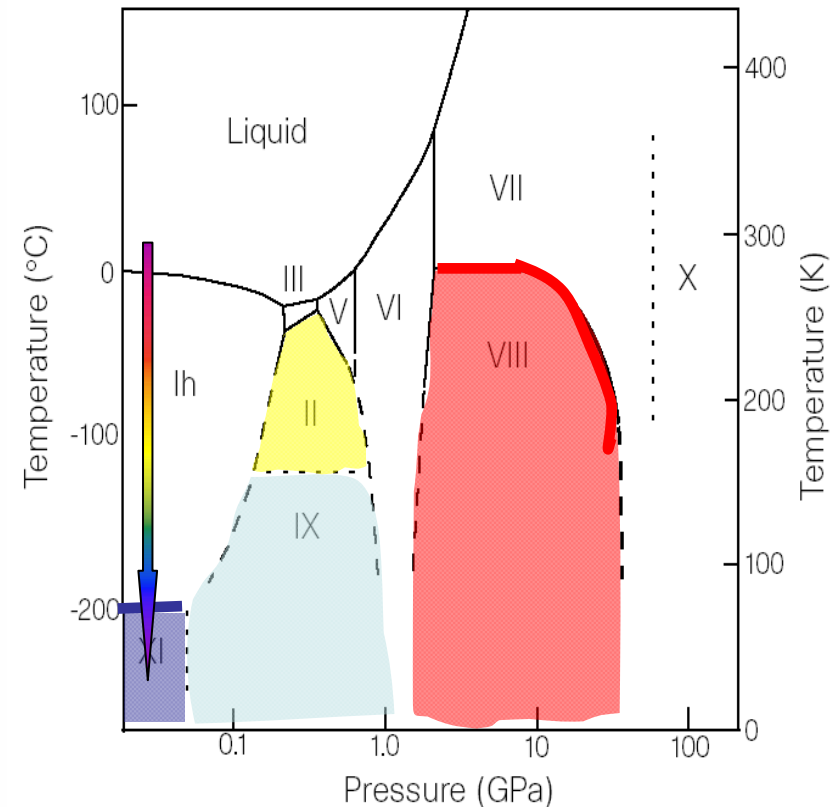
$$\Omega \propto (3/2)^N, S = k \ln \Omega = 3.37 \text{ J / K}$$



W.F. Giauque and J.W. Stout,
JACS 58, 1144 (1936)

$$S(T \rightarrow 0) = 3.4 \pm 0.6 \text{ J / K}$$

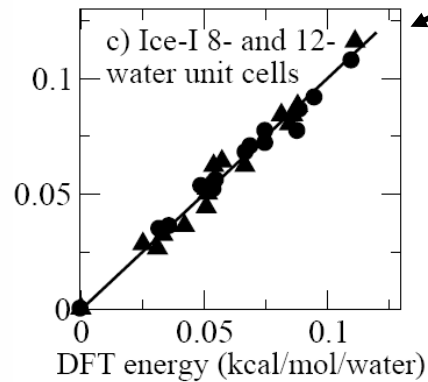
- Not just ice-Ih, many phases of ice are proton-disordered.
- Thermal properties of ice is essential to many important issues in physics, chemistry, and environment.
- Finding the proton-ordered ice, will help us better understand H-bonding.



Nature, vol 391, p.268

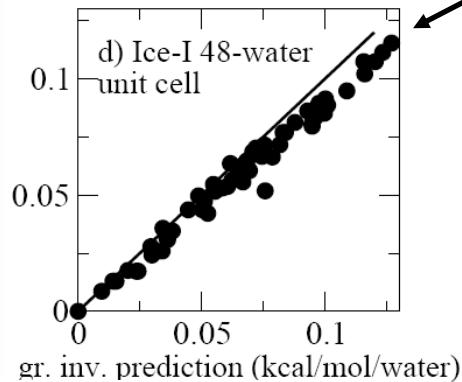
Proton Order/Disorder transition

Training graph invariants on small unit cells (*Quantum Mech.*)

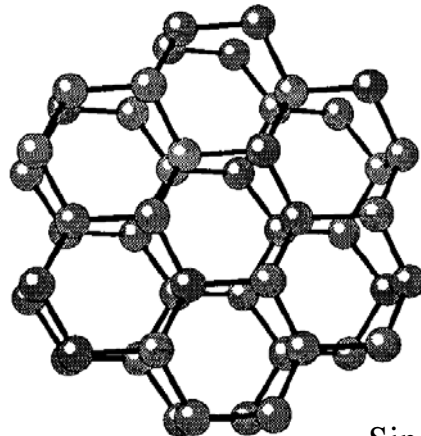
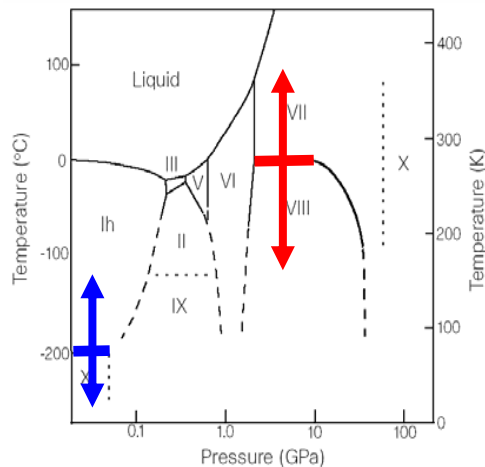
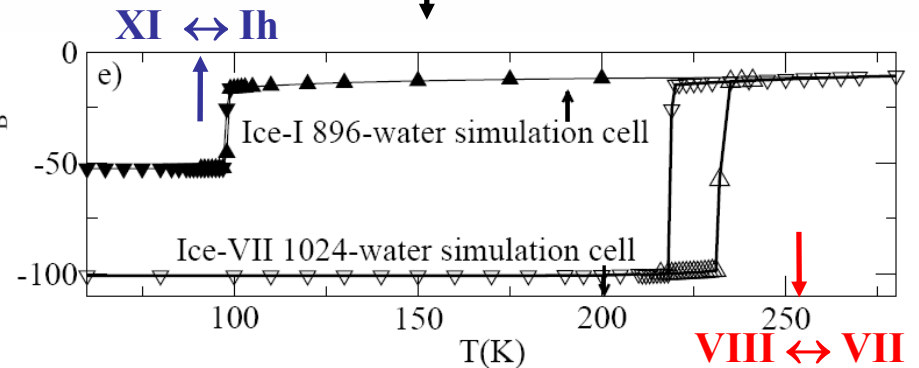


$$E(b_1, b_2, \dots) = E_0 + \sum_r \alpha_r I_r + \sum_{rs} \alpha_{rs} I_{rs} + \dots$$

Testing the prediction on larger unit cells. (*ab initio*) - QM



MC simulations on very large unit cells to simulate disorder/order transition. - SM

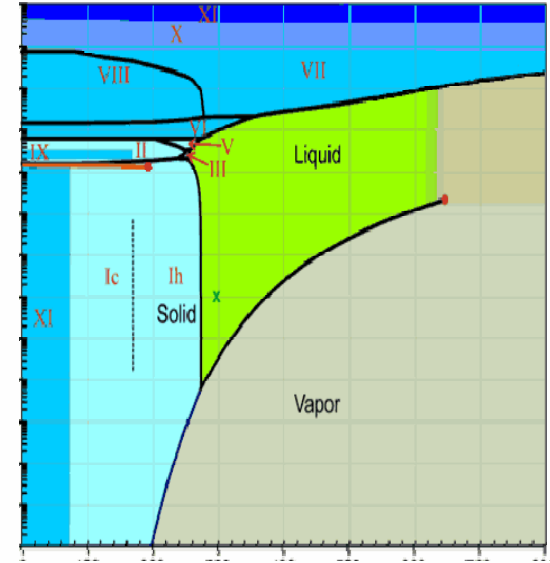


Singer, Kuo, *et al.*, *Phys. Rev. Lett.* **94**, 135702 (2005)]

What else about ice:

- *New Phases?*

- *Effects of Pressure?*



- *X-ray Absorption Spectra*



Zhang Jingyun

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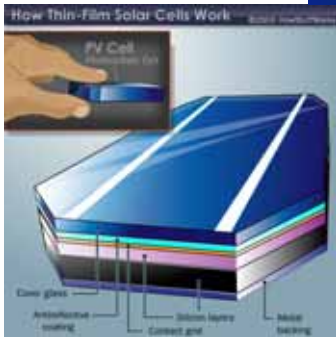
ZnO Alloys for opto-electronic applications



ZnO



LED

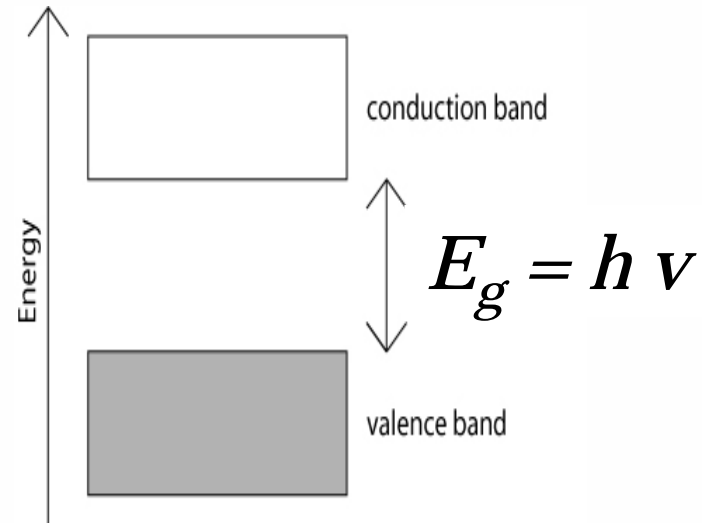


Solar Cells



Sun cream

➤ Band Gaps and light



➤ Tunable E_g

ZnO

$$E_{ZnO} = h v_1$$



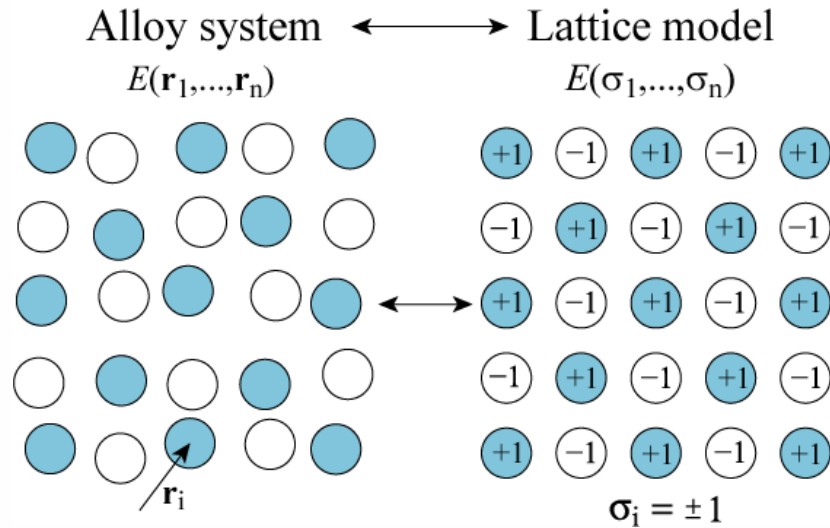
MgO

$$E_{MgO} = h v_2$$



Semiconductor Alloys – $Be_xZn_{1-x}O$

➤ Cluster Expansion Theory

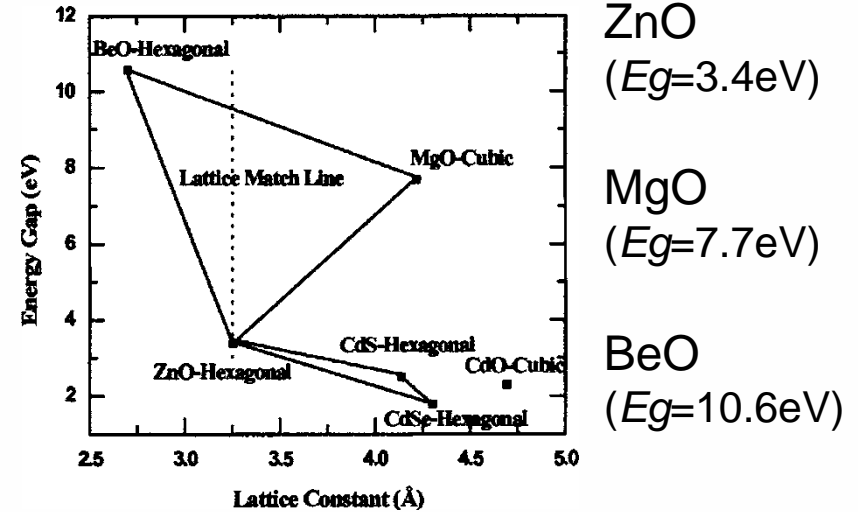


$$E(\sigma_1, \dots, \sigma_n) = \sum_{\{ij\}} J_{ij} \sigma_i \sigma_j + \sum_{\{ij,k\}} J_{ijk} \sigma_i \sigma_j \sigma_k + \dots$$

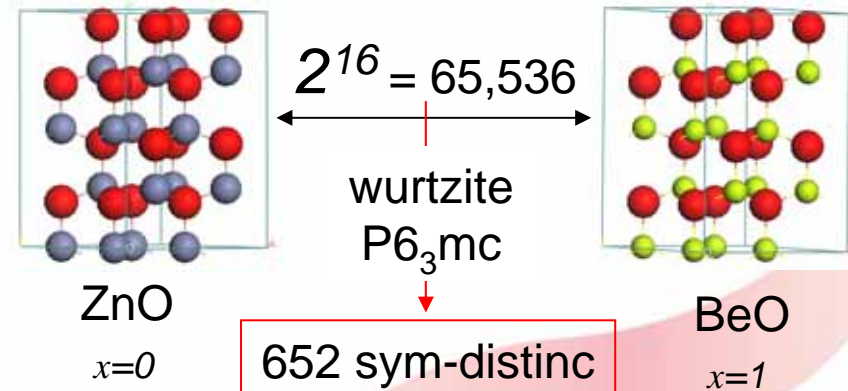
How many J_{ij} are needed? 2^N

How many σ are needed? 2^N

➤ $Be_xZn_{1-x}O$



APL **88**, 052103 (2006)



$\sigma = (\text{Zn}, \text{Zn}, \dots)$

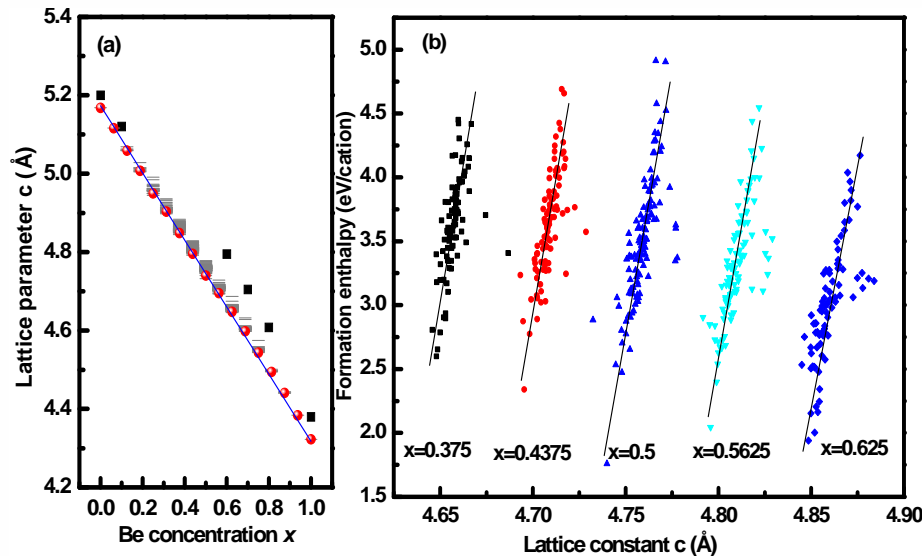
$\sigma = (\text{Be}, \text{Be}, \dots)$

$Be_xZn_{1-x}O$ Alloys

➤ Formation Enthalpy:

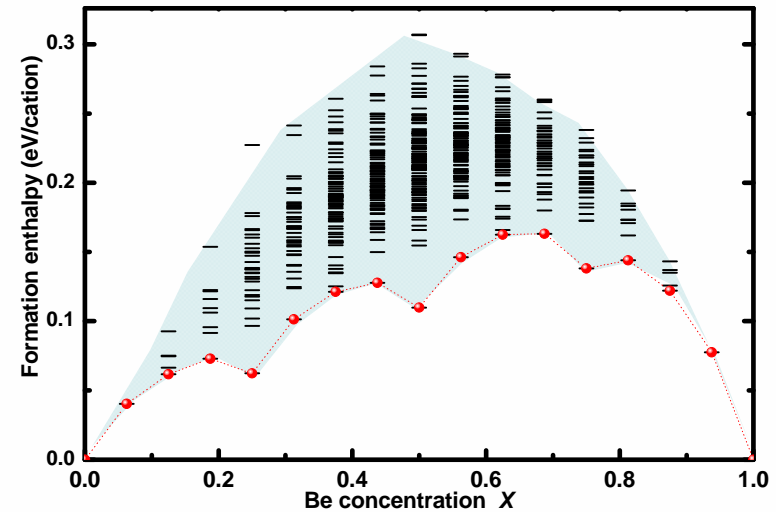
$$\Delta H_f[\sigma, x] = E[\sigma, Be_xZn_{1-x}O] - xE[BeO] - (1-x)E[ZnO]$$

➤ Lattice constants

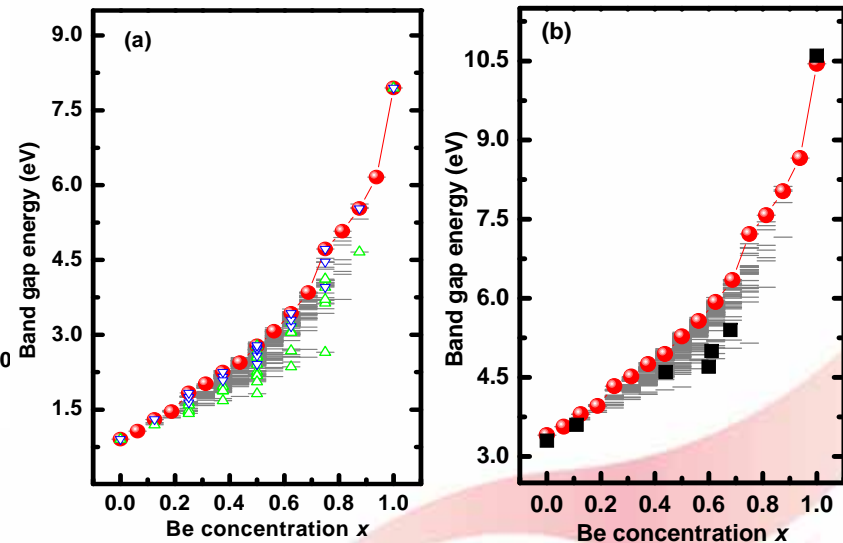


XF Fan, et al., and Jer-Lai Kuo, *App. Phys. Lett.* **91**, 121121 (2007)

Fan, Zhu et al, *APL*, **91**, 121121 ('07).



➤ Band Gaps

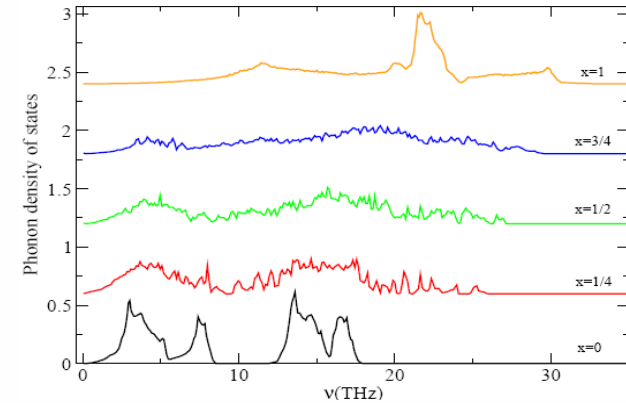


▲ [100] SL ▼ [001] SL ■ Exp

Effect of Lattice Vibration

➤ Phonon DOS

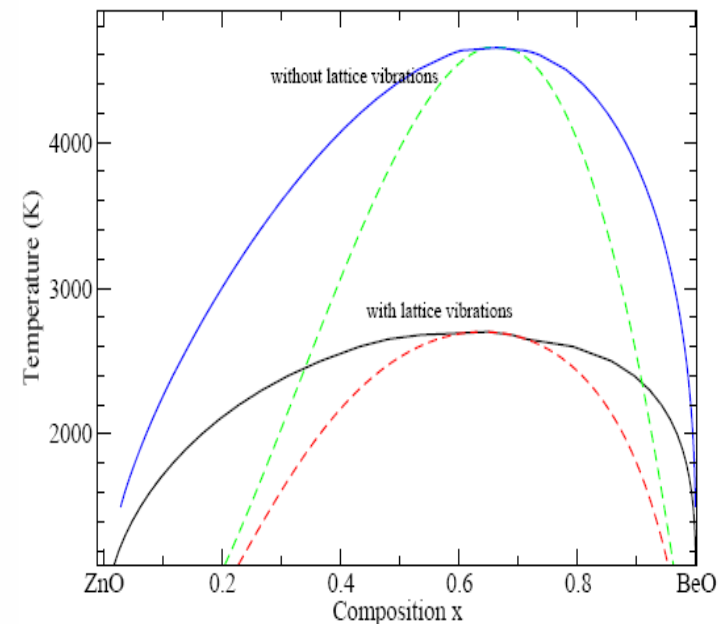
- PWSCF: linear response good for crystal.
- PHONE: calculate DOS



➤ Effect of Lattice Vibration

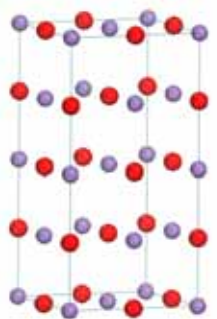
Approximations made:

- Bragg-Williams app.
- Neglect short range order
- Harmonic app.
- Small super-cell size

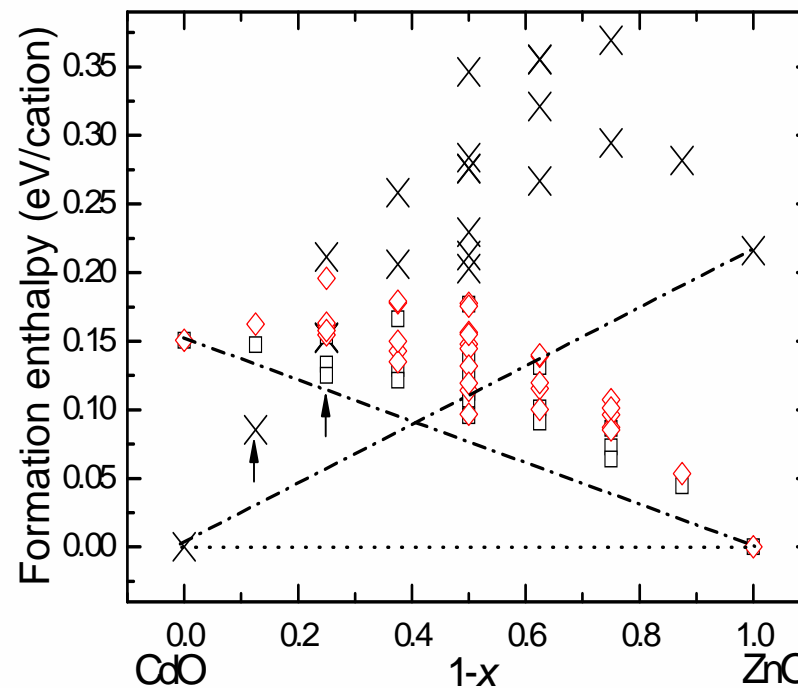
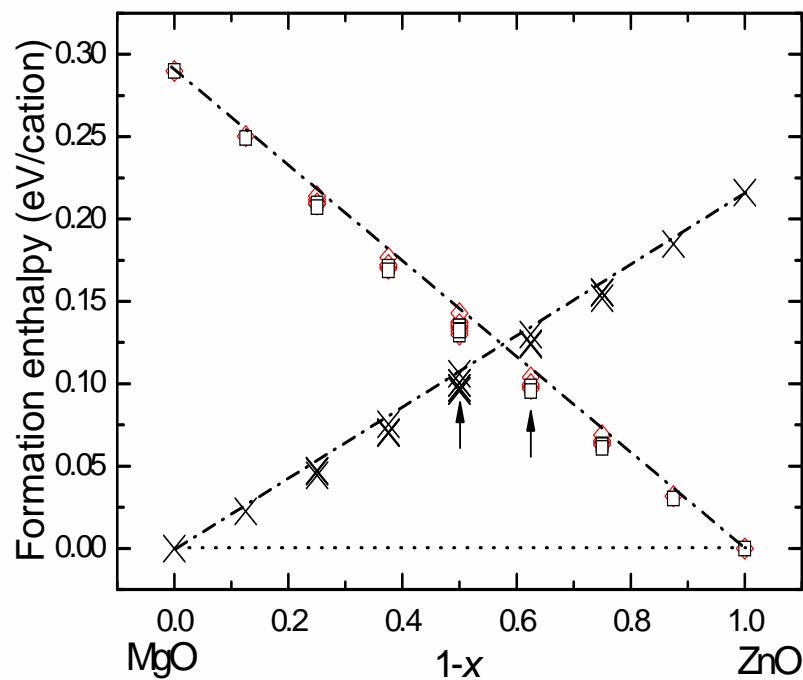
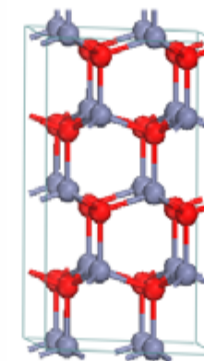


CK Gan, XF Fan, and Jer-Lai Kuo,
Comp. Mat. Sci. (in press)

Non-isostructural alloys



➤ $Mg_xZn_{1-x}O$, $Cd_xZn_{1-x}O$, 66



中央研究院

原子與分子科學研究所

INSTITUTE OF ATOMIC AND MOLECULAR SCIENCE
ACADEMIA SINICA, TAIPEI, TAIWAN

X. F. Fan, HD Sun, Z.X. Shen, and Jer-Lai Kuo,

J. Phys. Cond. Matt. **20**, 235221 (2008)



Material Discovery on the CLOUD

- 1st-Principle Calculations to replace wet-lab based methods?



- Improve efficiency

2 month (2007) with in-house PC clusters

2 days (2008) with in-house PC clusters

2 hrs (2009) with large HPC?

- Material Design



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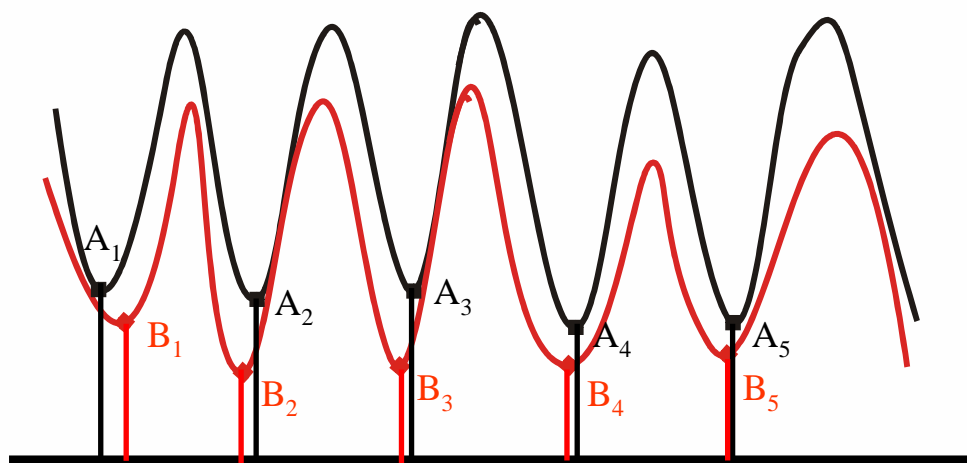
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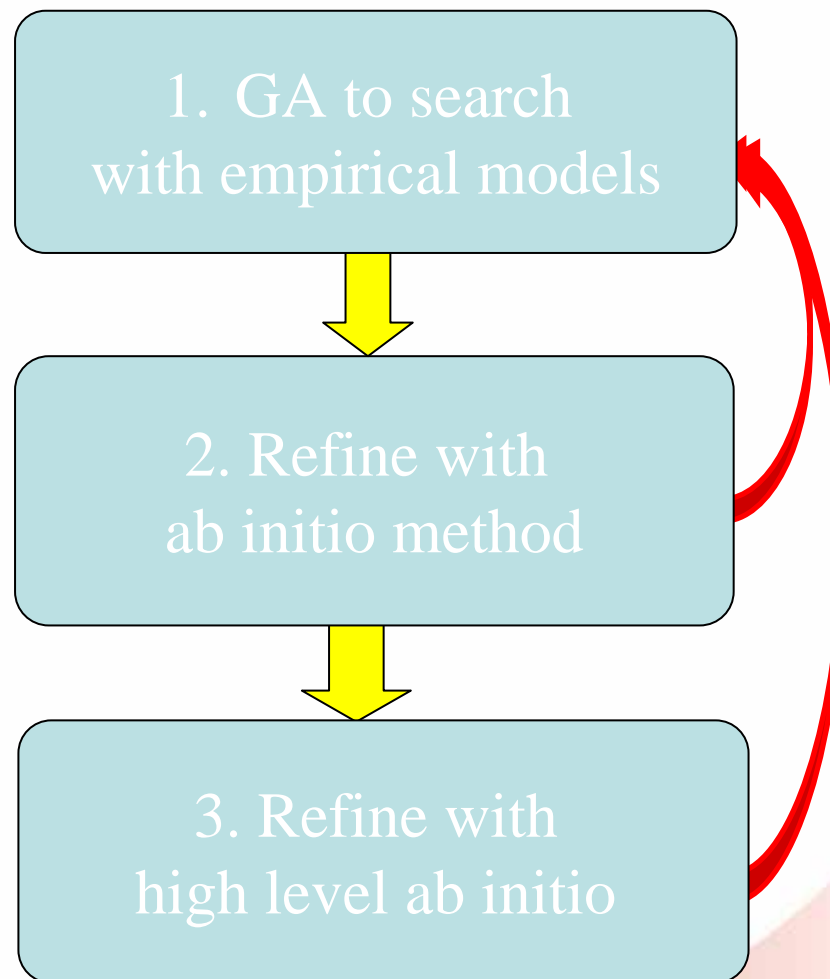
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Basic scheme of the Multi-scale Method

- Use empirical models to quickly explore the PES of water
- Ab initio results can be used in return to reparameterized models

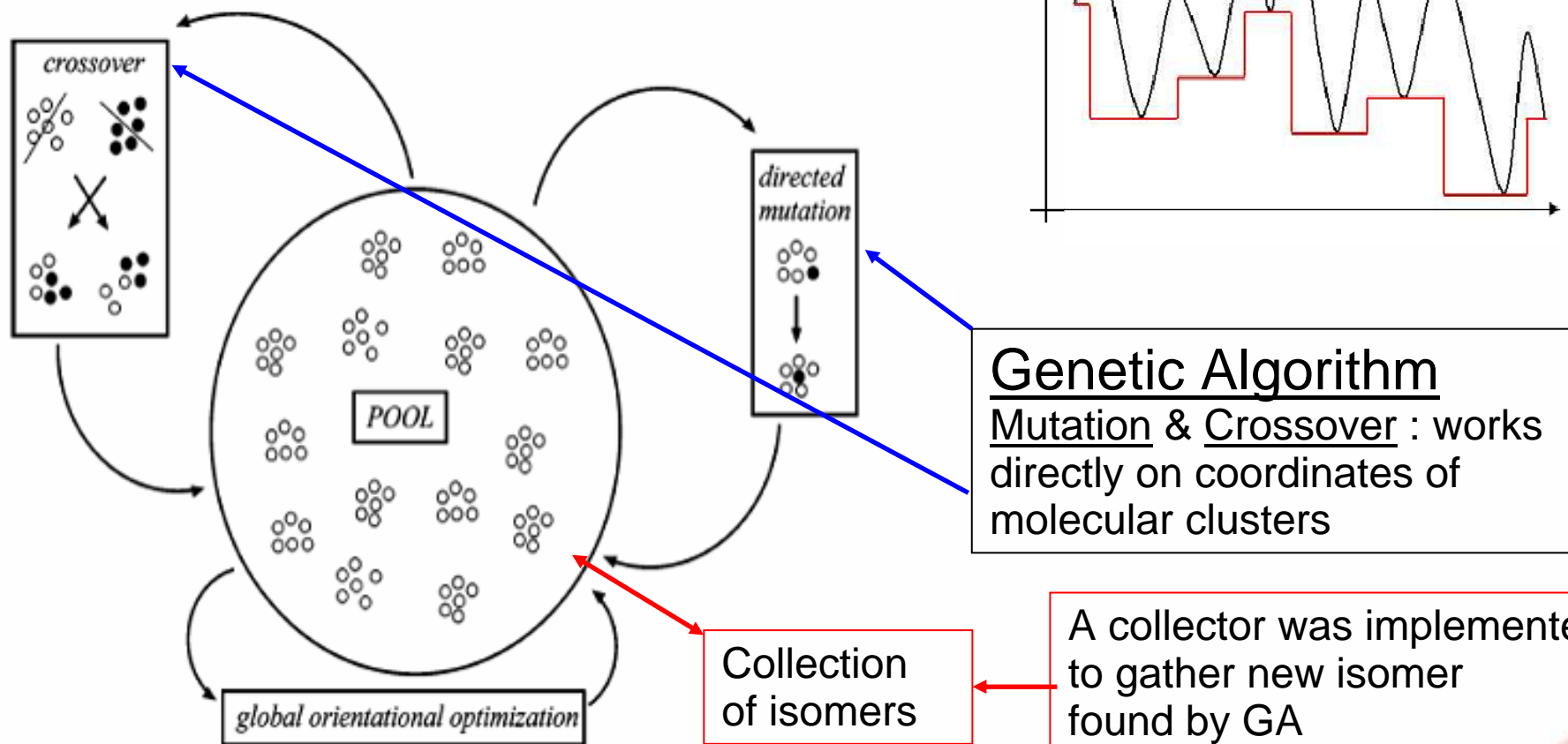


→ A replaced by (→ B → A)



Exploring the Energy Landscape of Clusters using Genetic Algorithms

Asynchronized GA: by Bernd Hartke

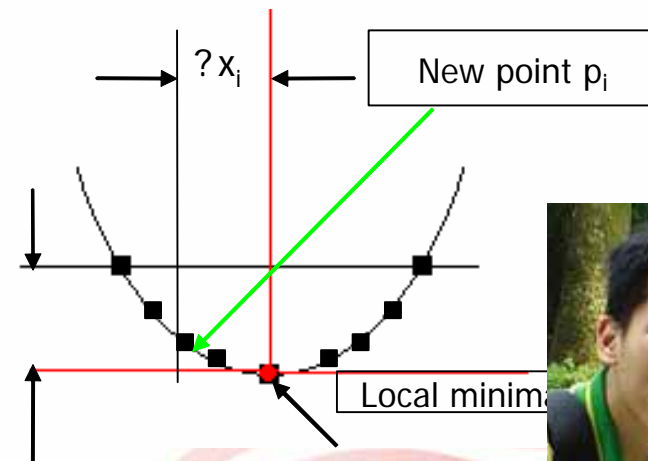
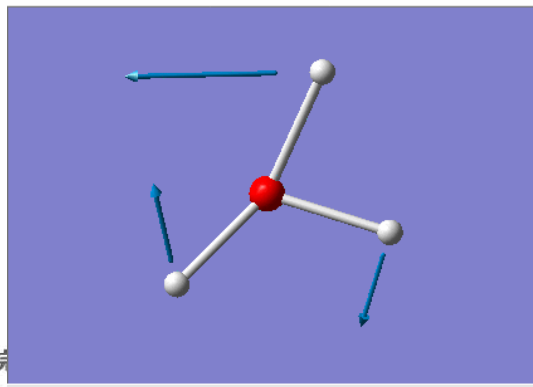
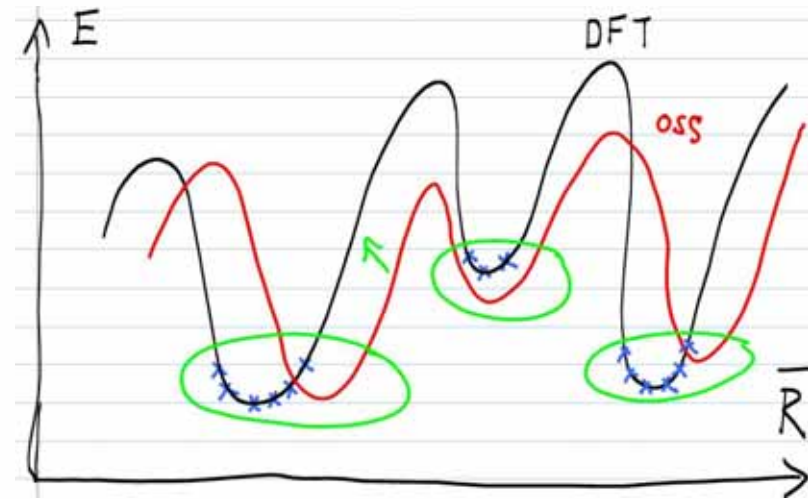


Parameterizing empirical models

- Levenberg-Marquardt nonlinear least square algorithm + Genetic Algorithm
- Objective function

$$F(p) = \sqrt{\frac{1}{M} \sum_{k=1}^M (E_{OSS2}^k(p) - E_{DFT}^k)^2}$$

p: parameter, M: # of data points,
 E_{OSS2}^k, E_{DFT}^k : OSS2 and DFT binding energy of configuration k

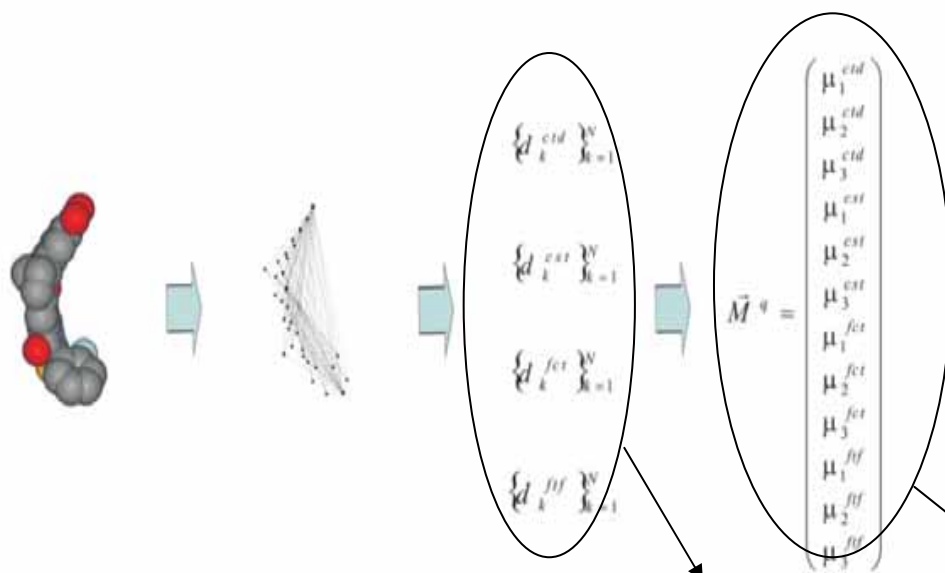
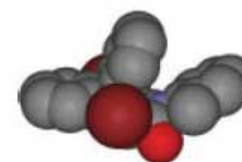


How to distinguish diff. isomers?

“Ultra fast shape recognition”

Ballester and Richards, J. Comp. Chem. **28**, 1711 (2007)

- Similarity index ranges from 0 to 1:
 - 0 : totally different
 - 1 : exactly the same
- Differ by just 1 atom, SI= 0.966



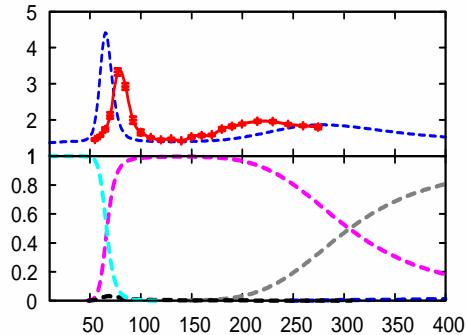
$\{d_k^{ctd}\}$: to molecular centroid (ctd)
 $\{d_k^{cst}\}$: to the closest atom (cst) of ctd
 $\{dkfct\}$: to the furthest atom (fct) of ctd
 $\{dkfff\}$: to the furthest atom (fff) of fct

To make them independent to the size of cluster, the first, second and third moments of each set are used.

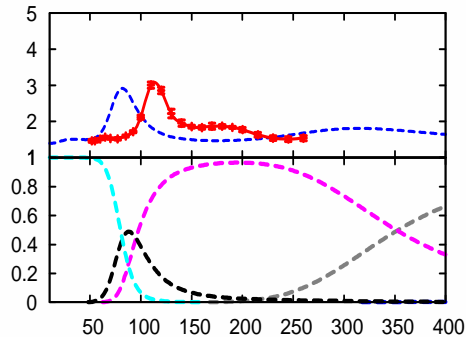


HSA: from isomers to thermal prop.

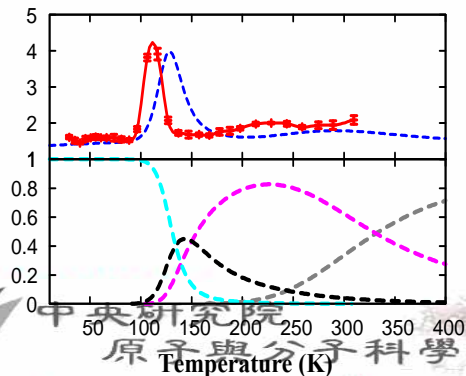
d) n=8



e) n=9



f) n=10



T --- L --- SR --- DR --- MR ---

Harmonic superposition approximation

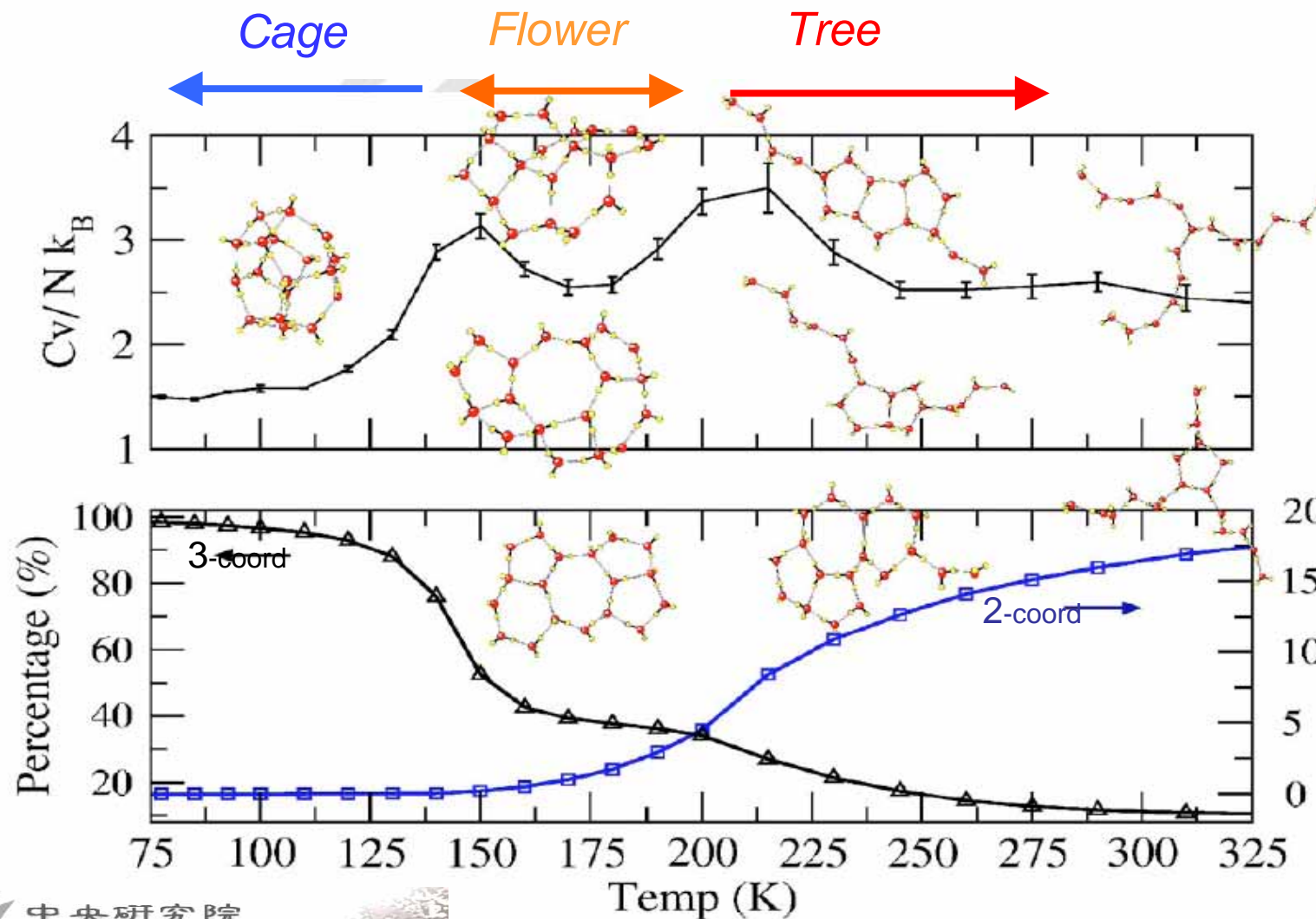
□ The canonical partition function $Z(\beta)$ can be approximated as the summation of harmonic contributions of all collected local minima.

$$\square Z(\beta) = \sum_a n_a Z_a(\beta)$$

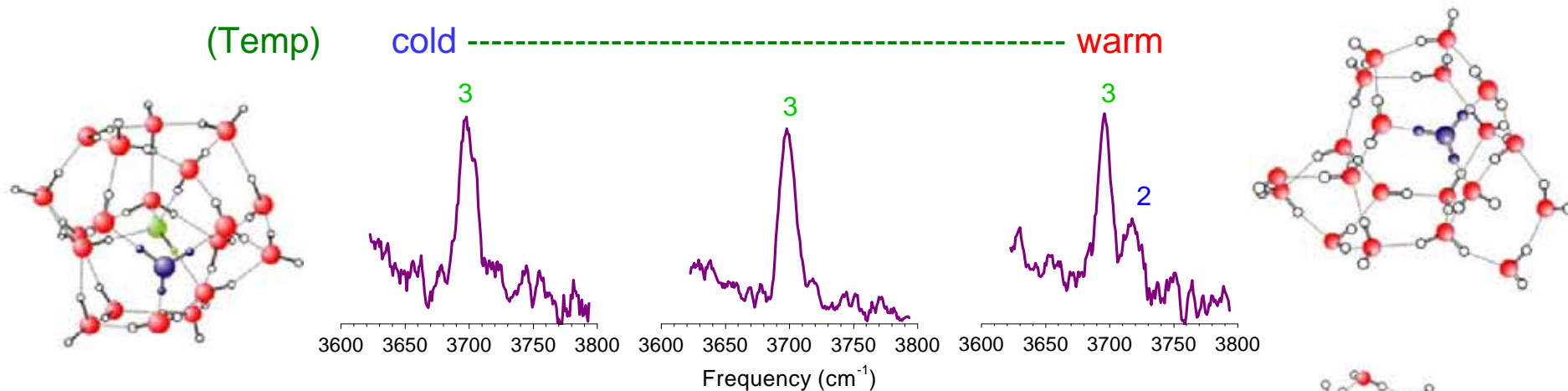
□ The finite temperature behavior (heat capacity curves, structural transitions, canonical probabilities .i.e.) of W_n^+ clusters can be derived afterward.

Nguyen, Ong, and Kuo,
"A multi-scale approach to study thermal behavior of protonated water clusters $H^+(H_2O)_n^+$ ",
J. Chem. Theory and Comp.)

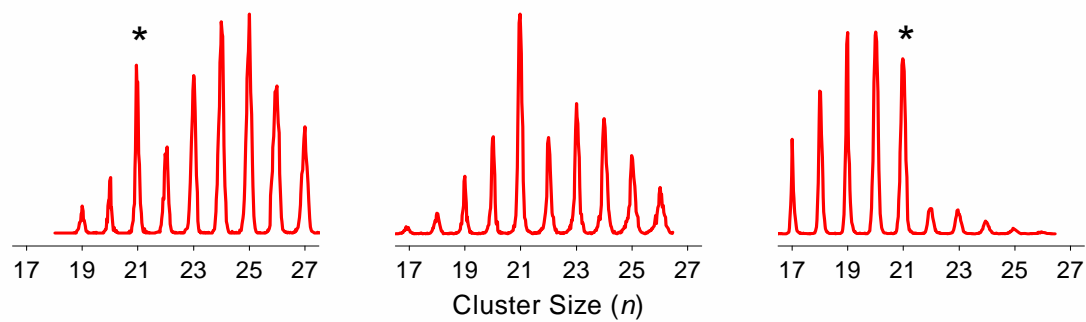
Two-stage melting of H^+ (H_2O)₁₈



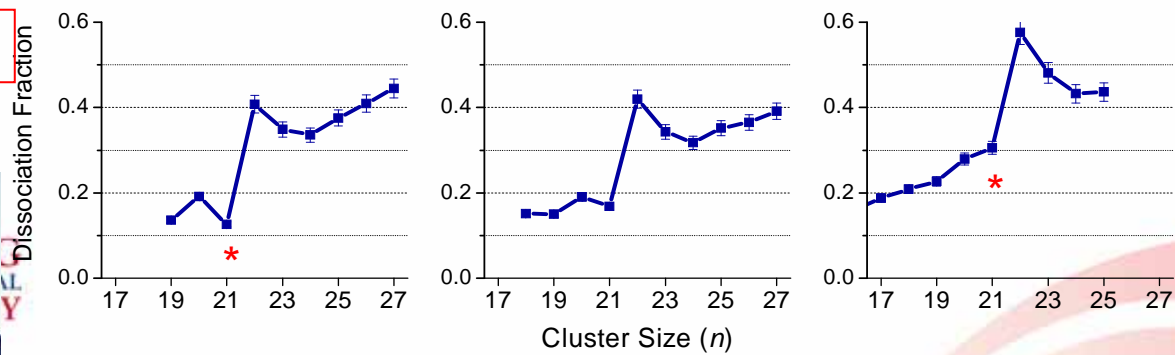
Melting of $H^+ (H_2O)_{21}$ cage (135~155K)



Mass spectra



Dissociation



Collaborators and Group Members



Prof. Mike Klein
Chem@UPenn



Prof. Ong Yewsoon
SCE@NTU



Prof. Zhu Zexuan
ShenZhen U/China

\$\$ =



Fan XiaoFeng



Wu Hongyu



Nguyen Quoc Chinh



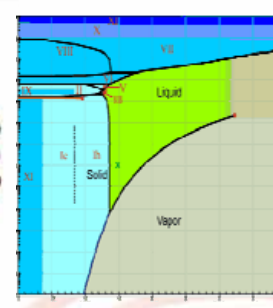
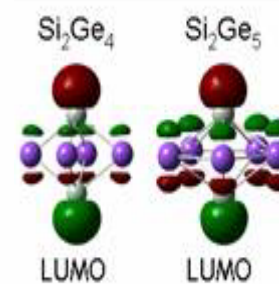
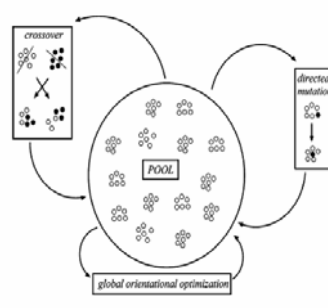
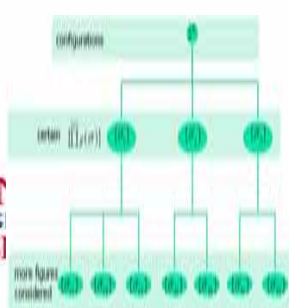
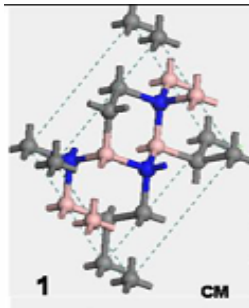
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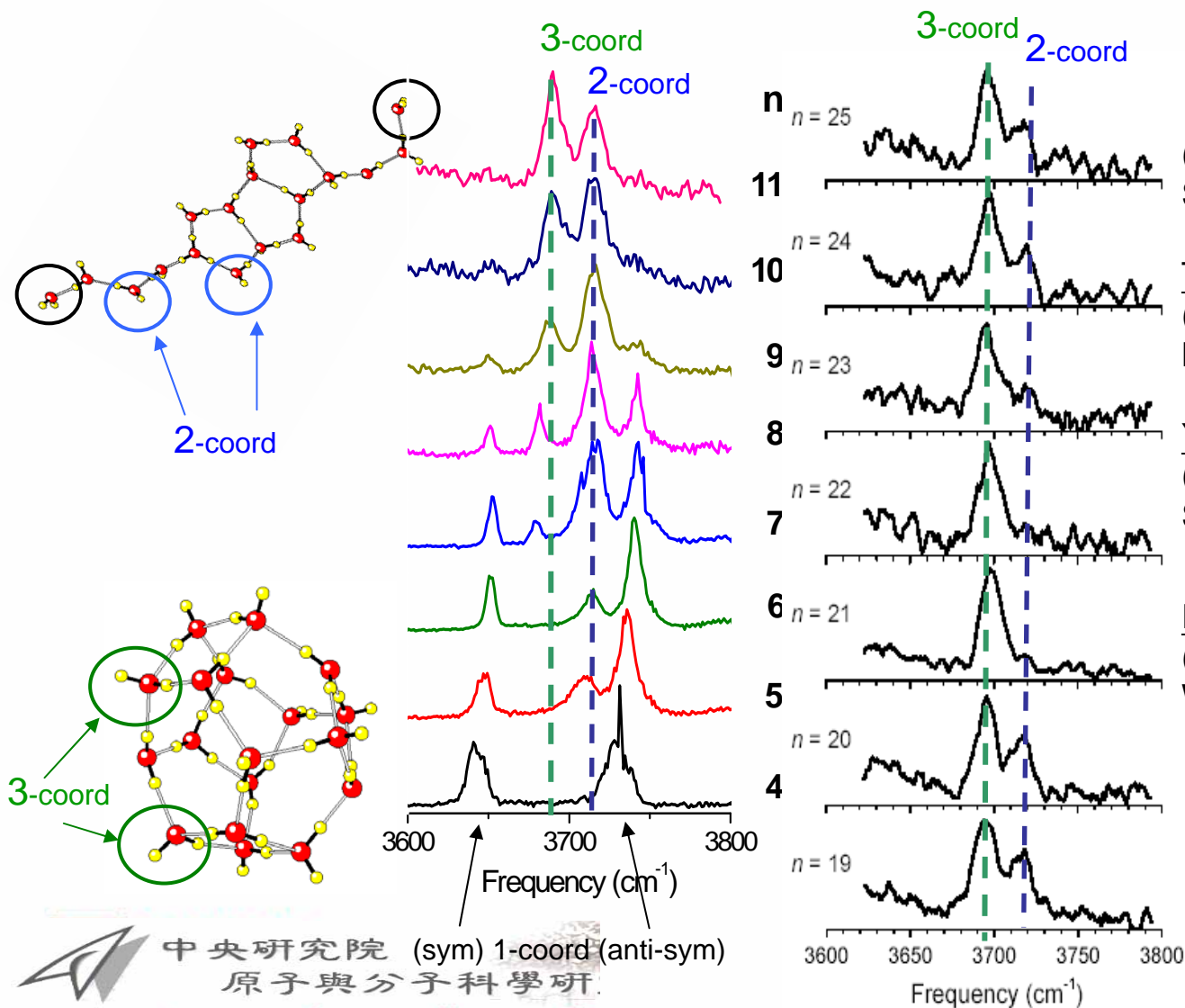
Zhang Jingyun



Lee Ching-Tao



Structural info from VPS



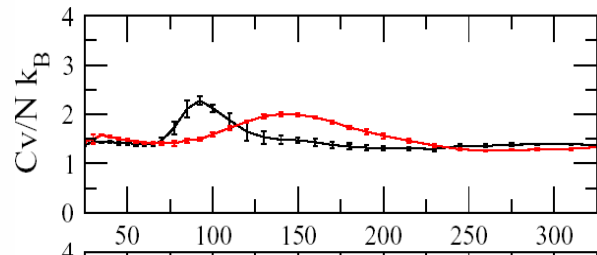
VPS:
(Vibrational Predissociation Spectra)

Tohoku U.
(Prof. Mikami and Prof. Fujii)
Miyazaki *et al.*, *Science* (04)

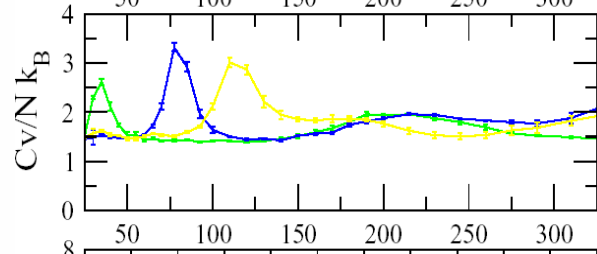
Yale-Georgia
(Prof. Johnson and Prof. Duncan)
Shin *et al.*, *Science* (04),

IAMS, Sinica
(Prof. Yuan Lee and Prof. Chang)
Wu, Lin, *et al.*, *JCP*, *PCCP* (05)

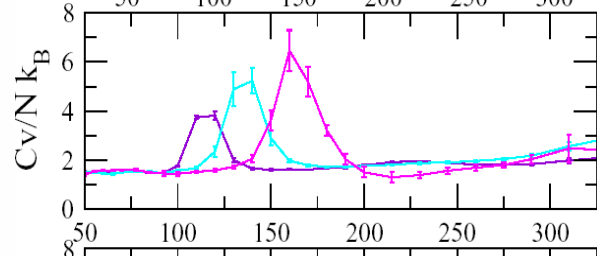
Thermal behavior of $H^+(H_2O)_n$



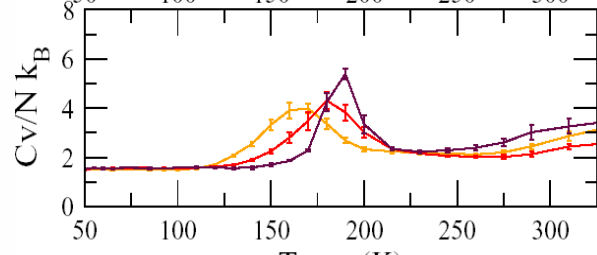
n=5
n=6



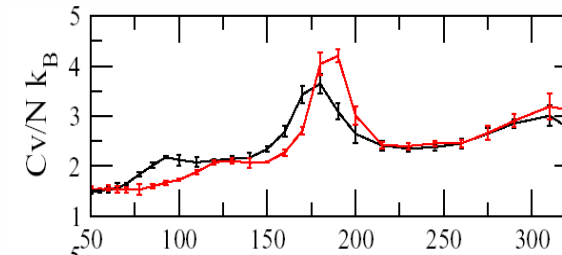
n=7
n=8
n=9



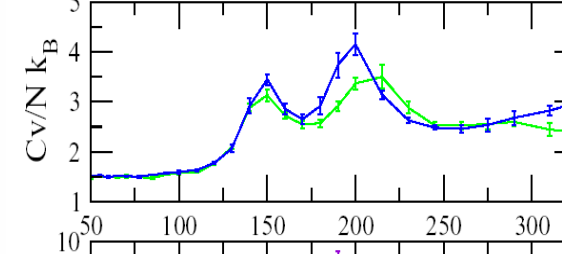
n=10
n=11
n=12



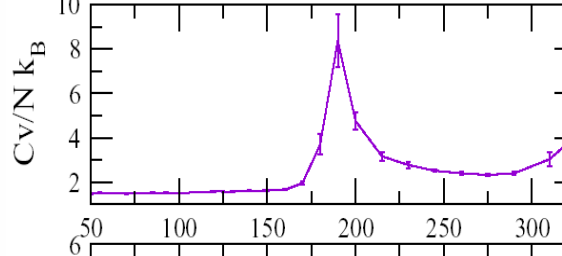
n=13
n=14
n=15



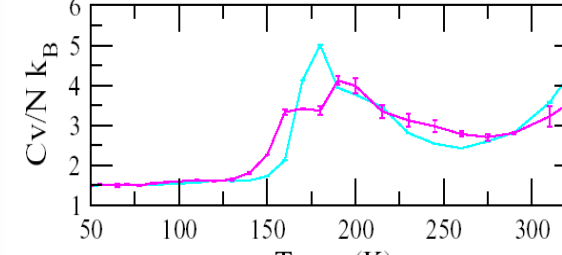
n=16
n=17



n=18
n=19



n=20



n=21
n=22

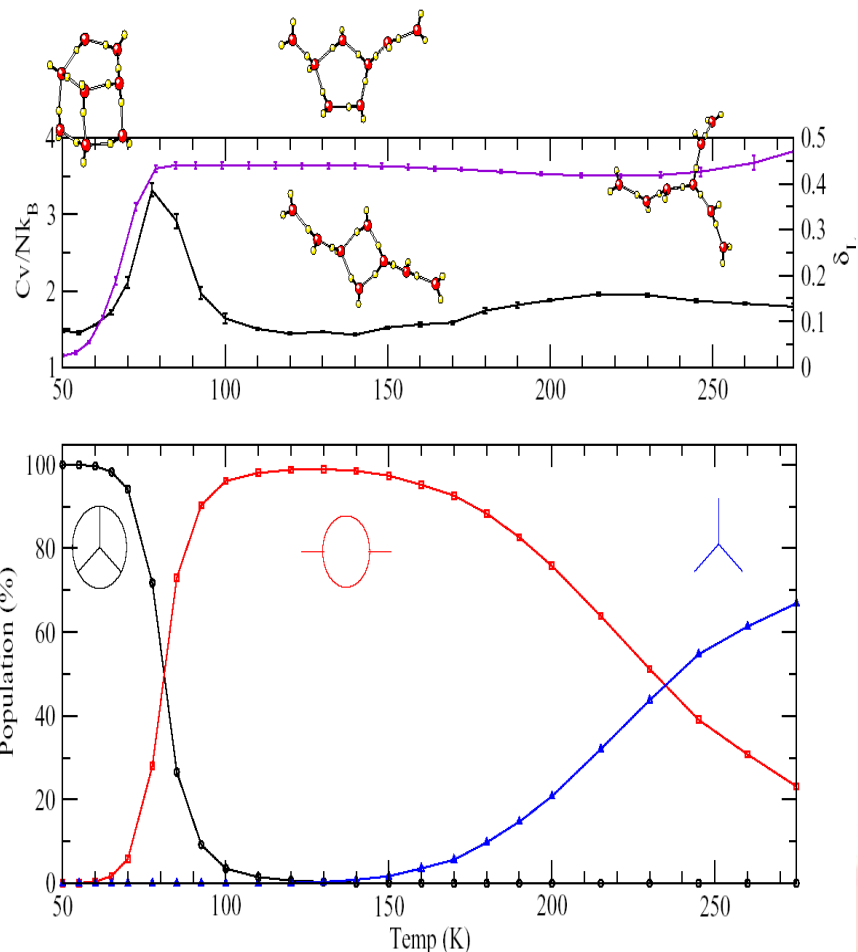
!! Peaks in C_v are associated to structural changes. !!

of Local minimum and why is statistical average important!

$(\text{H}_2\text{O})_n$

n	GA TTM2F	re-opt B6	SHS*
4	14	10	
5	33	23	
6	154	95	
7	639	406	
8	2331	1429	(166)*

(*) K. Ohno, J. Phys. Chem. A, 111, 10732 (2007)

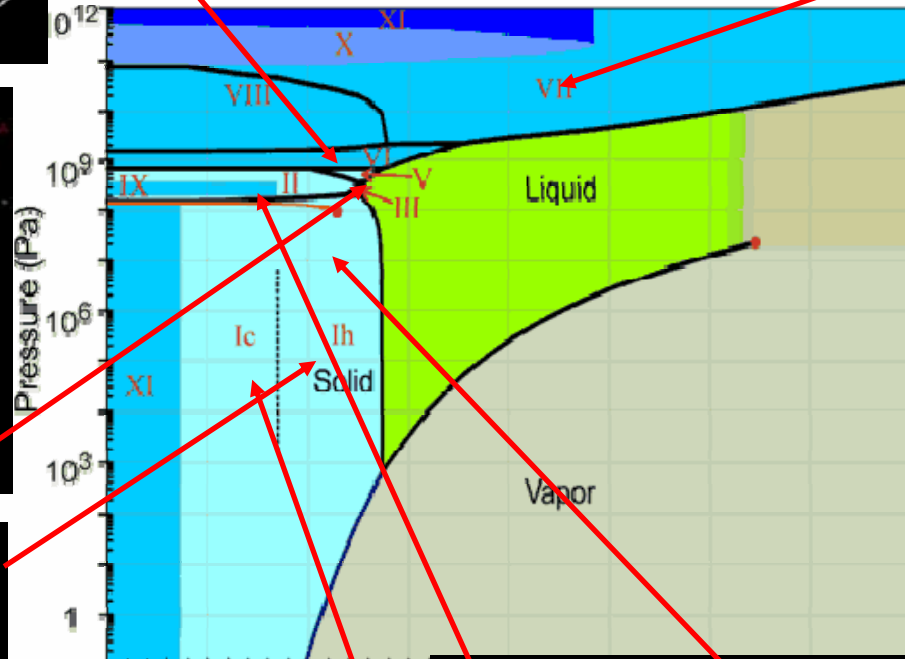
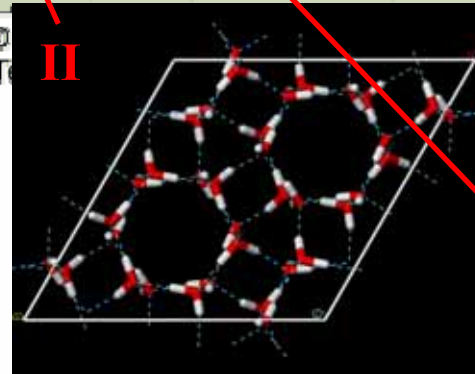
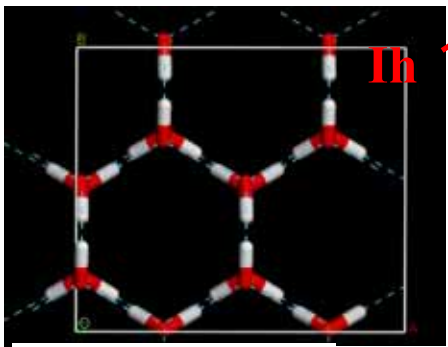
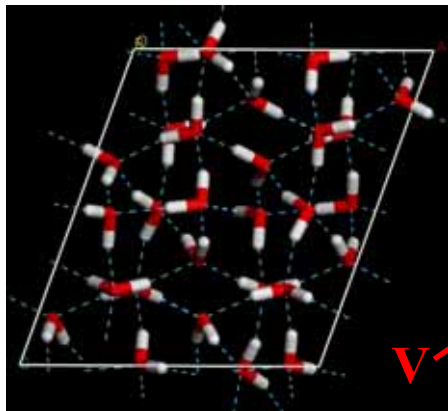
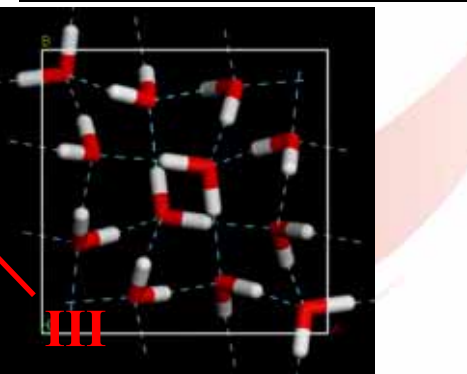
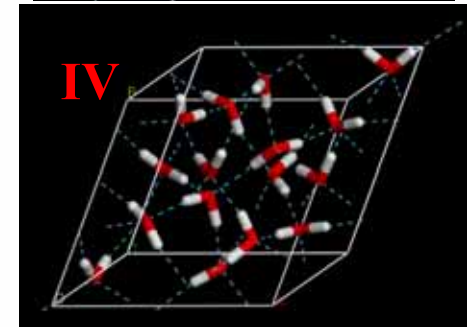
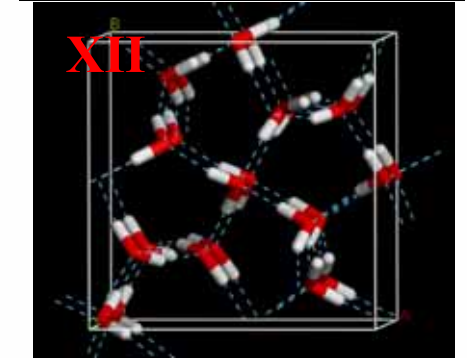
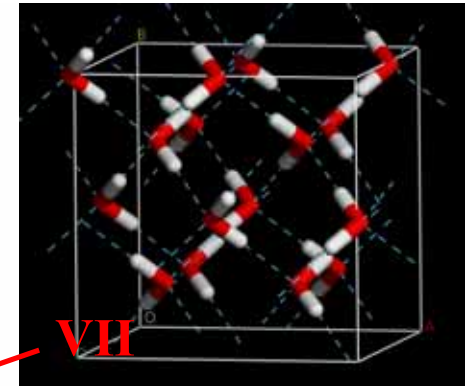
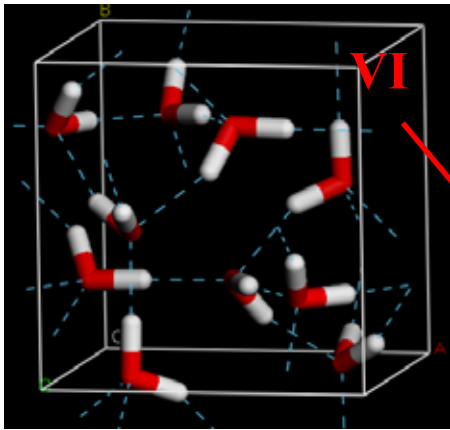


Dominating structures are the minimum of Free Energy!!

Development of Water Models

- OSS2 potential
 - Designed for $\text{H}^+(\text{H}_2\text{O})_n$
 - $V_{\text{total}} = V_{\text{charge interaction}} + V_{\text{polarization interaction}} + V_{\text{O-H interaction}} + V_{\text{O-O interaction}} + V_{\text{three-body interaction}}$
 - Polarizable
 - No artificial charge
 - Dissociable → Important in ionic chemistry
 - Overlap of electron clouds modelled by some screening functions





Be_xZn_{1-x}O Alloys : Phase Diagram

➤ Composite-T Phase Diagram

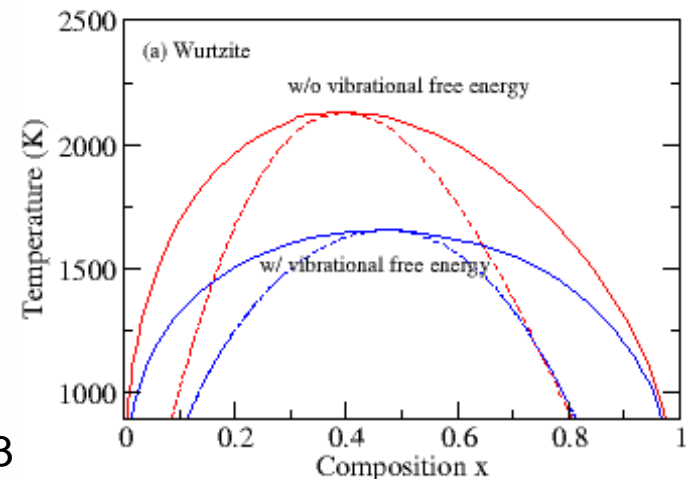
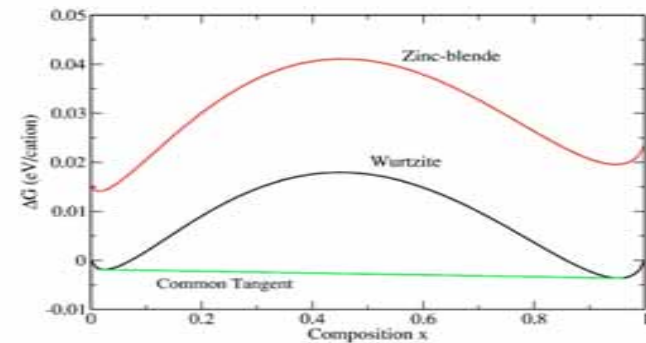
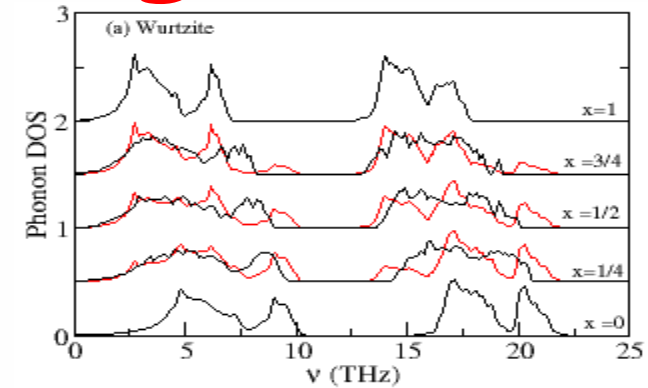
$$\Delta G = \Delta H_f - T \Delta S_{\text{conf}} + \Delta G_{\text{vib}}$$

- ΔH_f (formation enthalpy)
- ΔS_{config} : Bragg-Williams app. for random alloys
- ΔG_{vib} : including ZPE

➤ Effect of Lattice Vibration

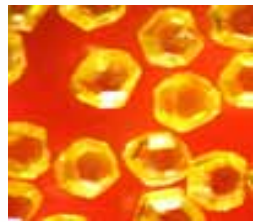
- PWSCF: linear response good for crystal.
- SQS: (special quasi-random structure)* to represent random alloy

▪ PHONON: calculate DOS



* Zunger, Wei, Ferreira and Bernard, PRL 65 (1990) 353

cubic-BC_xN: 2nd hardest material ??



diamond

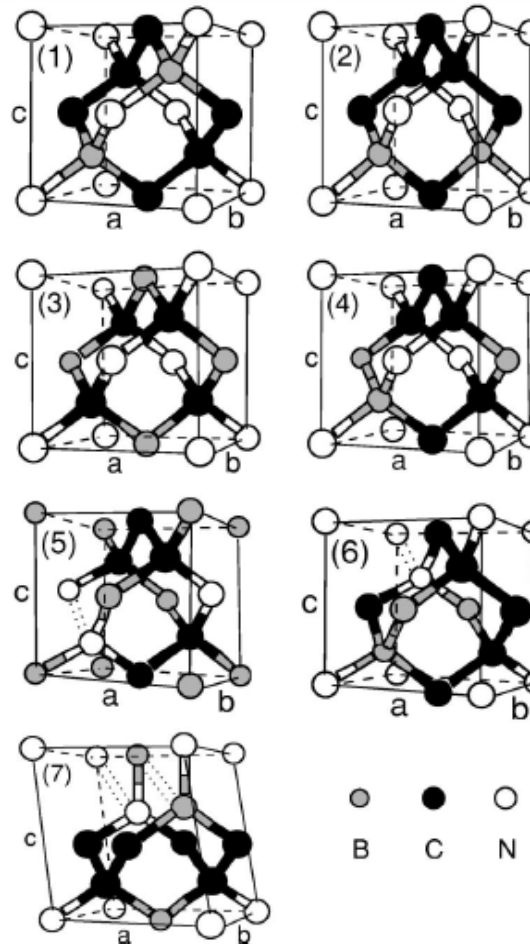
+



c-BN

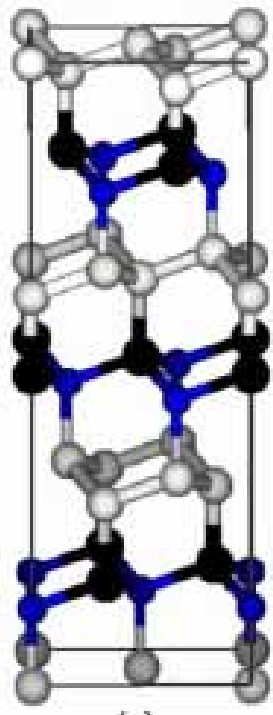
=??

Sun, ...Louie, **PRB**, **64**, 094108 (2001)



420 alloy config.
(8-atom cubic cell)

7 sym. dist. config.
(8-atom cubic cell)



Chen, Gong, and Wei,
PRL. **98**, 015502 (2007)

* (111) super-lattices
config. are more
stable

* They are also harder
than c-BN



cubic-BC_xN: Bond Counting Rules

